

Determination of Structural and Possible Ferromagnetic Properties of $Cr_xSn_{1-x}O_2$ Compounds:

First Principle Approach

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ABSTRACT: The total energy calculations of structural and magnetic properties of a six atom supercell of pure, binary, tetragonal structure SnO_2 at $x = 0$ and twelve atom supercells of ternary transition metal oxides $Cr_xSn_{1-x}O_2$ at $x = 0.25, 0.50, 0.75$ and 1.00 were simulated using first-principles calculations within the framework of generalized gradient approximation (GGA) for spintronic utilities. Doping Cr with wide band gap SnO_2 has the effect of transition from a non-metal to metallic state. The calculated magnetic moment at $x = 0.25$ is $1.9976 \mu B$, $x = 0.50$ is $3.9309 \mu B$, $x = 0.75$ is $5.8831 \mu B$ and $x = 1.00$ is $7.8271 \mu B$. The room temperature ferromagnetism attained in this study is apparent in Sn substitution with Cr atom.

KEYWORDS: Ferromagnetic, Diluted semiconductor, Metal oxides, Spintronic devices.

1. INTRODUCTION

Dilute magnetic semiconductor is a non-magnetic semiconductor doped with magnetic atoms most especially transition metals to induce magnetic and spin-polarized materials which exhibits both ferromagnetism and semiconductor properties. It's of interest because of its functionality in the development of spintronic devices. The substituted Cr^{2+} ions has magnetic moment attribute into Sn^{2+} ions in $Cr_xSn_{1-x}O_2$ at various compositions; $x = 0.25, 0.50, 0.75$ and 1.00 respectively without having any negative consequence on the resulting structure of CrO_2 at $x = 1.00$.

Tin oxide (SnO_2) is a wide band gap semiconductor oxide and is useful in magnetic data storage and resonance imaging, photo catalyst (Pearson et al., 2003), increase refractoriness (Jednak et al., 2011), electrodes and anti-reflection coatings in solar cells (Wolf et al., 2001), manufacturing of gas sensors, optoelectronic devices and resistor (Al-Saadi et al., 2019), polishing powder, glass coatings and making of liquid crystal display and so on.

Chromium(IV) oxide (CrO_2) is metastable at room temperature and pressure and is of interest as a result of its potential usage in spintronic heterostructures, superconductor structures (Anwar et al, 2010), electronic storage device, catalyst and corrosion inhibition devices, magnetic heads and magnetic field sensors (Bate, 1978) polishing agents medicine, (Kurmaev et al., 2003) to mention a few. Transition metal oxide, SnO_2 doped with Cr atom finds usefulness in production of riboflavin biosensors (Lavanya, 2013), processing of pigments, gas sensors and optional applications. The spin and orbital magnetic moments of Cr and O in CrO_2 using local spin density approximation (LSDA) approach was reported (Jeng and Guo, 2002). The computed sub-lattice magnetic moment of Cr in rutile-type CrO_2 of $2.04 \mu B$ was deducted (Huang et al., 2018).

Ferromagnetism entails magnetic ordering in which the intrinsic magnetic dipole moment or spin of electrons on each crystal-lattice site all align in the same direction. The two unique features of ferromagnetic are: spontaneous

magnetization of which the total magnetization exists within a uniformly magnetized microscopic volume without the presence of a field and magnetic ordering temperature. Ferromagnets either possess strong attractive or repulsive forces in the presence of permanent magnet (Aibangbee and Onohaebi, 2018). Utilization of ferromagnetism are: data storage, electromagnets, magnetic tape recording and transformers (Hummel, 2013).

There have been considerable studies of Cr doped semiconductor, SnO₂ both experimental and theoretical wise (Stashans et al., 2014, Kuppan et al., 2017, Mishra et al., 2015) with various methodologies (Abidi et al., 2013, Leite et al., 2005 and Kasar et al., 2013). Albeit, there is need to study the effects of chromium doped semiconductor material SnO₂ at x=0.25, 0.50, 0.75 and 1.00 using PBE-PAW technique and density functional theory based so as to understand its structural and room temperature magnetic properties as an effective gas sensor on ceramic materials.

2. COMPUTATIONAL DETAILS AND THEORETICAL BACKGROUND

The appraisals of this study were actualized by using Perdew-Burke- Ernzhof Projected Augmented wave within the framework of Generalized Gradient Approximation (GGA) based density functional theory and Perdew-Burke- Ernzhof Projected Augmented Wave (PBE- PAW) technique (Ayedun et al., 2017). The PBE- PAW approach is preferred in this work because it enhances adequate and accurate facts about equilibrium state, structural minimization, response, spectroscopic and magnetic properties of crystalline structures and so on. Also, PBE- PAW technique (Perdew et al., 1997) is time and cost effective, accurate pseudopotential libraries are highly available and accessible in the Quantum Espresso package (Giannozzi et al., 2009). Cr_x Sn_{1-x} O₂ retains its crystalline tetragonal structure from compositions x = 0 to 1. The details of volume (V)-energy (E) were fixed in the second –order Birch-Murnaghan equation of state (Birch, 1994; Murnaghan, 1994) and minimized lattice parameters a and c (since a = b, but b ≠ c) at room temperature were examined.

$$\delta E(V) = E - E_0 = BV_0 \left[\frac{V_n}{B'} \right] + \left(\frac{1}{(1-B')} \right) + \left(\frac{V_n}{B'(B'-1)} \right) \dots\dots\dots (1)$$

The plane wave cut-off energy is set at 80 Ry for a K- mesh of 10 x 10 x 5 for a pure six atom supercell, SnO₂. The Monkhorst – Pack (Monkhorst and Park, 1976) K-meshes of 10 x 10 x 5, 8 x 8 x 4, 10 x 10 x 5 and 10 x 10 x 5 for a twelve atom supercell built and kinetic energy of 90 eV each were set at x = 0.25 to 1.00 respectively. Self consistent relaxation of atomic positions were quantumly simulated and the optimized energy converged to 0.01 milli electronVolt (meV) at each concentration x in respect of Brillouin zone and cut – off energy. Comparison is made with existing experimental and theoretical works via calculations performed in non-defective SnO₂ at x = 0 and doped ternary compounds of Cr_x Sn_{1-x} O₂ at x = 0.25 to 1.00, with minimized structural parameters of a and c as indicated in Table 1.

Table 1: Optical lattice parameters of $Cr_x Sn_{1-x} O_2$ compound.

Parameter(Å)	Present Study	Existing Works	
		Experimental	Theoretical
a	4.7358	4.743, 4.7202, 4.7471	4.8959, 4.780
c	3.1853	3.2070, 3.200	3.30371, 3.268

The values of optimized lattice parameters enlisted in Table 1 are in conformity with existing experimental and theoretical studies (ESI, 2013).

3. RESULTS AND DISCUSSION

3.1 Structural Properties

The pattern of undoped (six atoms supercell), binary rutile SnO₂ and doped ternary transition metal oxides (twelve atoms supercell) $Cr_x Sn_{1-x} O_2$ at concentrations $x = 0.25$ to 1.00 remains tetragonal rutile body centered cubic structure. The details of volume (V) – Energy (E) were imputed in the second order Birch- Murnaghan equation of state (Birch, 1947; Murnaghan, 1947) and minimized lattice parameter a, bulk modulus B and pressure derivative B' at room temperature were determined. Vegard's law examined the rules of mixture using lattice parameter of a solid solution of two constituents A and B at constant temperature and uniform crystal structure. This law is not limited to only pure binary solid solution A and B, it is also applicable to ternary transition metallic oxide, $Cr_x Sn_{1-x} O_2$. The lattice parameter as attested by Vergard's law is:

$$a(A_x B_{1-x} C) = x a_{AC} + (1 - x) a_{BC}, \dots\dots\dots (2)$$

herein, a_{AC} and a_{BC} are the equilibrium lattice constants of CrO₂, SnO₂ and $a(A_x B_{1-x} C)$ is the ternary compound lattice constant, x is the molar fraction of Cr in SnO₂ for $0 < x < 1$. Because of lack of compliance to Vergard's law in semiconductors both theoretical (Ayedun, 2017, Abdiche, 2010) and experimental works (Gu et al., 2007; Wei et al., 2020;Savidan et al., 2010) can now be written as:

$$a(A_x B_{1-x} C) = x a_{AC} + (1 - x) a_{BC} - x (1 - x) b, \dots\dots\dots (3)$$

where b stands for bowing constant. The substitution of Sn with Cr atoms at $x = 1$, results into mini overvalue of lattice constant and a better transition compound produced. The lattice parameter of undiluted SnO₂ at $x = 0$ (4.7358 Å) and that of alloyed transition metal oxide at $x = 1$ (4.6442 Å) are very close as revealed in Table2.

Table 2: Lattice parameter a, bulk modulus B and pressure derivative B' of $Cr_x Sn_{1-x} O_2$ compound

Nomenclature	Parameters	Present Study	Experimental Values	Theoretical Data
SnO ₂	A (Å) B (GPa) B' (GPa)	4.7358 413.0 9.63	4.743, 4.7202277 2.0	4.8959, 4.780312
Cr _{0.25} Sn _{0.75} O ₂	A (Å) B (GPa) B' (GPa)	4.7834 577.1 15.0	-	-
Cr _{0.50} Sn _{0.50} O ₂	A (Å) B (GPa) B' (GPa)	4.7777 663.2 15.0	-	-
Cr _{0.75} Sn _{0.25} O ₂	A (Å) B (GPa) B' (GPa)	4.7305 907.7 14.48	-	-
CrO ₂	A (Å) B (GPa) B' (GPa)	4.6442 660.1 3.62	4.421 239	4.384, 4.459 237.7

The component of chromium compositions modification with lattice constant and bulk modulus were considered. The calculated optimized lattice constants were plotted against chromium compositions(x) in Figure 1. As the chromium dopant increases from x = 0 to 1.00, the downward bowing constant of ternary compound $Cr_x Sn_{1-x} O_2$ is -0.34. The difference in lattice constant at x = 1(4.6442 Å) compared with that of existing theoretical data (4.384 Å and 4.459Å)by Srivastav et al., 2008 and Huang et al., 2018 respectively, depicted in Table 2 is as a result of GGA artefact used.

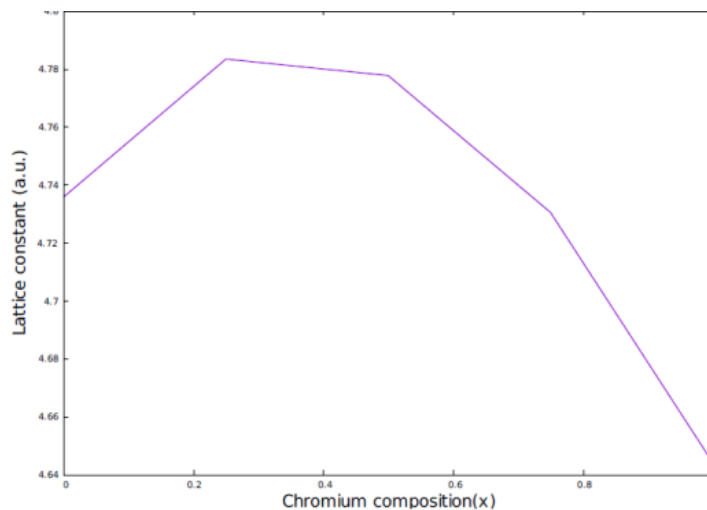


Figure 1: Lattice constant of $Cr_x Sn_{1-x} O_2$ as function of chromium composition (x)

The bulk modulus varies with rising chromium composition from x = 0.25 to 0.75 and decrease to 660.1 GPa at x = 1.00 as indicated in Figure 2. The increase in bulk modulus is due to the fact that GGA has tendency to overestimate and the bulk modulus bowing constant is -0.90.

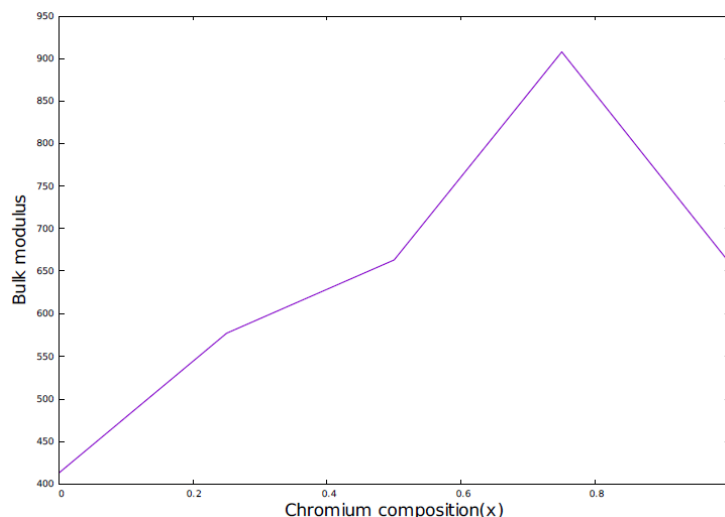


Figure 2: Bulk modulus of $Cr_x Sn_{1-x} O_2$ as function of chromium composition (x)

3.2Magnetic Properties

Magnet has the capacity to create magnetic field. Magnetic field is produced as much as orbits are set in motion as well as there exist the spinning of electrons. The three major types of magnet are electromagnet, permanent magnet and temporary magnet. The feature of magnet comprises of its attractive property, directive property, pair property to mention a few. Magnets are grouped into diamagnetic, paramagnetic, ferromagnetic, ferrimagnetic antiferromagnetic. The magnitude of magnetic moment which quantifies the magnetic properties of material is a function of an unpaired electron. At $x = 0$, SnO₂ is diamagnetic because both Sn²⁺ and O²⁻ are paired and the net magnetic field effect is zero (Wang et al., 2010). The outer valence electrons; $Sn(4d^{10} 5s^2 5p^2)$, $Cr(3s^2 3p^6 3d^4)$, $O(2s^2 2p^4)$ which are unpaired enhanced increase in magnetic moment as dopant atom, chromium increases from $x = 0.25$ to 1.00 . The bulk magnetic moment of $Cr_x Sn_{1-x} O_2$ compounds were examined at $x = 0$ to $x = 1.00$ as shown in Table3.

Table 3: Magnetic moment of $Cr_x Sn_{1-x} O_2$ compound in μB .

Composition (x)	Nomenclature	Present study	Experimental	Theoretical
0	SnO ₂	0	0	0
0.25	Cr _{0.25} Sn _{0.75} O ₂	1.9976	-	-
0.50	Cr _{0.50} Sn _{0.50} O ₂	3.9309	-	-
0.75	Cr _{0.75} Sn _{0.25} O ₂	5.8831	-	-
1.00	CrO ₂	7.8271	-	9.0

Magnetic moment increased linearly with risen chromium atom substitution as displayed in Figure 3. A weak ferromagnetic ($1.9976 \mu B$) was observed at $x = 0.25$ and a strong ferromagnetic material emerged at $x = 1(7.8271 \mu B)$.

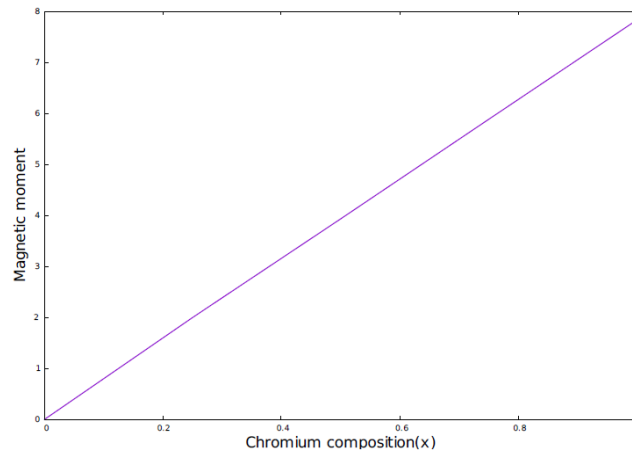


Figure 3: Magnetic moment of $Cr_x Sn_{1-x} O_2$ as function of chromium composition (x)

4. CONCLUSION

Cr atom was substituted into $Cr_x Sn_{1-x} O_2$ ternary compound at various composition $x = 0$ to 1.00 using PBE-PAW approach. The addition of the dopant increased the lattice parameter, enhanced the pressure as well as the bulk modulus. It influenced the phase transition from nonmagnetic material SnO_2 to room temperature ferromagnetic system at $x = 0.25$ to 1.00 without distortion to its tetragonal rutile structure. The researchers are recommended to experimentally work on the bulk magnetic moment of SnO_2 with transition metals using various methodologies.

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